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**Research Article** 

# Novel ZnCo<sub>2</sub>O<sub>4</sub>/WO<sub>3</sub> nanocomposite as the counter electrode for dye-sensitized solar cells (DSSCs): study of electrocatalytic activity and charge transfer properties

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#### ABSTRACT

Nowadays, Pt coated FTO is used conventionally as the counter electrode in dye-sensitized solar cell (DSSC). In addition to the high price of the Pt electrode, it reduces the stability of the DSSC. In this study, we introduce and study a new counter electrode based on the  $ZnCo_2O_4/WO_3$  composite that is used in DSSC. We show that the efficiency of the DSSC can be enhanced even more then the Pt-based one by employing the  $ZnCo_2O_4/WO_3$  as counter electrode. By examining the structural, morphological, optical, and electrochemical properties of the synthesized electrodes, we investigate the counter electrodes synthesized under different conditions. The XRD patterns and FESEM images confirm that the composite phase of the  $ZnCo_2O_4/WO_3$  layers is formed. Additionally, electrochemical studies by CV, Tafel, EIS, and Mott-Schottky methods indicate the electrocatalytic activity of the  $ZnCo_2O_4/WO_3$  sample have significantly increased compared to  $ZnCo_2O_4$  and  $WO_3$  electrodes. Furthermore, the characterization of DSSCs with TiO<sub>2</sub> photoanode and different counter electrodes show that the efficiency of the solar cells based on  $ZnCo_2O_4/WO_3$  has a promising efficiency of 7.76%, which has increased by 7% compared to the Pt one.

#### 1. Introduction

Since the introduction of dye-sensitized solar cells (DSSCs) in 1991 by O'Regan and Gratzel [1], they have emerged as a potential alternative for thin-film solar cells. DSSCs have always been affordable due to their low manufacturing cost and non-toxicity, and have also attracted the consideration of environmentalists [2,3]. A typical DSSC consists of photoanode, counter electrode, electrolyte, and adsorptive dye molecules. In the last two decades, extensive research has been carried out on different parts of the DSSC. Investigating the possible structures for the photoanode and using different materials has been studied frequently [4,5]. Widespread studies have been done to replace conventional electrolytes such as  $I^-/I_3^-$  and  $Co^{3+}/Co^{2+}$  [6,7]. Also, many researchers have sought to find metal coordination complex or organic dye molecules instead of the conventional N719 [8,9]. However, in most cases, non-toxic and inexpensive TiO<sub>2</sub> as photoanode and N719, which has appropriate molecular level aligning with TiO<sub>2</sub>, have been the best option. Likewise, due to the proper compatibility of the redox levels of the  $I^-/I_3^-$  electrolyte with N719, their joint use is not far from expectation.

In recent years, replacing Pt as the counter electrode and getting rid of this expensive material has attracted the attention of studies [10-14]. The use of materials such as carbon black, graphene-based and

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carbon-based nanoparticles, PEDOT and even some conductive polymers such as PANI has raised hopes for replacing Pt. Narudin et al. [15], by using carbon black-graphite counter electrode enhance the efficiency of DSSC up to 5.74%. Kasi Reddy et al. [16], demonstrate the high electrocatalytic activity of bilayer PEDOT:PSS/SWCNH counter electrodes for achieving 5.1% power to current efficiency. Employing PANI counter electrodes, Karakuş et al. [17], succeeded to attain 6.3% efficiency for standard liquid electrolyte based DSSCs.

Using composite materials and taking advantage of the simultaneous properties of each component can be inspiring. Composite structures can be effective as they have demonstrated before [18]. Gao et al. [19], achieved 8.72% efficiency by synthesizing In<sub>4</sub>SnS<sub>8</sub>@MoS<sub>2</sub>@CNTs composite through hydrothermal method and using it as counter electrode. By replacing the standard Pt counter electrode with PANI/WSe<sub>2</sub> composite one, Sheela et al. [20], attained an 8.22% power to current efficiency, which was higher than that of the pure PANI and WSe<sub>2</sub> or even Pt counter electrodes. Yang et al. [13], elevated the electrocatalytic activity of CZTS counter electrodes by covering Co<sub>9</sub>S<sub>8</sub> on CZTS thin film prepared by the spin-coating method. This composite counter electrode demonstrated an improved efficiency of 6.41% in comparison with 3.92% efficiency of bare CZTS counter electrode. Nitrogen-decorated CeO<sub>2</sub>/reduced graphene oxide nanocomposite (CeO<sub>2</sub>/N-rGO) used as counter electrode in DSSC structure and the electrocatalytic activity for triiodide/iodide reduction been investigated by Wei et al., [21]. The DSSC fabricated based on CeO2/N-rGO demonstrated an advanced efficiency of 3.20%.

 $ZnCo_2O_4$  as an intrinsic p-type material benefits from high conductivity, structural stability, and high electrocatalytic activity [22]. It shows superior electrical conductivity and electrochemical activity than ZnO and  $Co_3O_4$  [23,24]. Meanwhile, it has low manufacturing price due to inexpensive and earth-abundant components, which guarantees the reduced cost-effective production [25]. By environmental friendliness, high electrochemical activity and conductivity,  $ZnCo_2O_4$  can be a proper candidate as counter electrode in DSSCs. Also, WO<sub>3</sub> with unique optical and electrochemical properties is another candidate for appropriate reduction of electrolyte species [26]. It also includes good physico-chemical and electrical properties, which can help to improve the electrocatalytic activity of the counter electrode along with the  $ZnCo_2O_4$  [27].

ZnCo<sub>2</sub>O<sub>4</sub> and WO<sub>3</sub> nanostructures have previously been used in DSSC structure as the counter electrode. Hou et al. [28] synthesized flower-like ZnCo<sub>2</sub>O<sub>4</sub> and graphene oxide nanostructures using solvothermal and common Hummers technique, respectively, and fabricated ZnCo<sub>2</sub>O<sub>4</sub>/RGO nanohybrids as counter electrode using hydrothermal method. Their DSSC based on this hybrid counter electrode has shown an efficiency of 7.22%. Wang et al. [29] have also achieved 6.73% efficiency for DSSC using ZnCo2O4/RGO composite as the counter electrode. Abdullaev et al. [30] have succeeded in producing core-shell ZnO@ZnCo<sub>2</sub>O<sub>4</sub> nanostructures that yielded 8.39% efficiency for a DSSC based on this counter electrode. By synthesizing ZnCo<sub>2</sub>O<sub>4</sub>@NiMoO<sub>4</sub> composite on carbon paper (CP) by two-step hydrothermal method, Zhang et al. [31] have succeeded in achieving 9.30% efficiency for DSSC with this counter electrode. ZnO@WO3 core-shell nanoparticles were prepared by Mahajan et al. [32] by sol-gel method, which yielded 5.73% efficiency for DSSC based on this counter electrode. Sulfurization treatment of mesoporous WO3/carbon film coated on fluorine-doped tin oxide (FTO) glass yielded WO3@WS2@carbon CE by Shen et al. [33]. Photovoltaic performance measurements have showed that the DSSC with the WO3@WS2@carbon core-shell counter electrode attained a power conversion efficiency of 7.71%. Tungsten trioxide was sprayed onto ITO conductive glass and filled with activated charcoal powder (ACP) for use as counter electrode in DSSC by Cui et al. [34]. The power conversion efficiency of WO<sub>3</sub>@ACP-based DSSC was 5.04%, which has been 3.15 times better than the 1.61% of WO3 DSSC.

In this research, by preparing a composite layer of ZnCo<sub>2</sub>O<sub>4</sub> (ZCO)

and  $WO_3$  materials and using it as the counter electrode in DSSC, we investigate its properties.  $WO_3$  structures can play an effective role in electron transport due to their proper conductivity. Simultaneously, ZCO nanoparticles can improve the performance of the counter electrode in combination with  $WO_3$  due to their proper electrocatalytic activity and charge transfer properties.

### 2. Experimental methods

#### 2.1. Synthesis of nanoparticles

A simple and rapid combustion method is used for the synthesis of ZCO nanoparticles. Primary, 1 g Zn(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and 2 g Co (NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O were dissolved in 8 g double distilled water, and while stirring vigorously, its temperature raised to 85 °C. Then, 6.5 g C<sub>6</sub>H<sub>8</sub>O<sub>7</sub> was added to the solution and the stirring continued for 15 min. The solution was transferred to an oven to heat at 300 °C for 20 min. After drying, the resulting powder was crushed in a mortar and then annealed for 5 h at a temperature of 600 °C with 5 °C/min rate in air furnace to obtain ZCO nanoparticles.

To synthesize the WO<sub>3</sub> nanostructure, sol-gel method was used. 1.5 g sodium tungstate dihydrate (Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O) were dissolved in 100 ml double distilled water and stirred for 10 min. Then, using 2 M HCl acid, the pH of the solution was reduced to 1.5. This solution was stirred for 11 h at room temperature to obtain a cloudy precipitate of WO<sub>3</sub>. The sediments were centrifuged several times and dried at 90 °C for 2 h. To improve the crystallinity, heat treatment was performed on the obtained powders at a temperature of 500 °C for 2 h.

To fabricate the ZCO/WO<sub>3</sub> composite material, solid state reaction method was used to react together the materials we synthesized in the previous steps. ZCO and WO<sub>3</sub> nanostructures with ratio of ZCO:WO<sub>3</sub> = 2:1were transferred into zirconium cups and milled in 50 ml pure ethanol for 4 h. After this procedure, the obtained solution was placed on hotplate at a temperature of 80 °C for a full day. After drying, the samples were crushed again with a mortar to be ready for deposition.

#### 2.2. Counter electrode fabrication

Drop-coating technique was employed to prepare the counter electrodes. In this method, slurry pastes of WO<sub>3</sub>, ZCO and ZCO/WO<sub>3</sub> nanostructures was prepared; 60 mg precursor was added to 1.3 ml pure ethanol and dispersed by ultrasonic device. 100  $\mu$ L of this solution was poured by a precise sampler onto FTO (1.5 × 1.5 cm<sup>2</sup>). To control the geometry of the counter electrode, a special ribbon was used around the FTO. Then, the samples were heated up to a temperature of 50 °C on the hotplate to finish the dispositioning operation.

#### 2.3. Photoanode preparation and DSSC assembling

Standard P25 TiO<sub>2</sub> was employed to fabricate the photoanodes. 100 mg P25 TiO<sub>2</sub> along with 120 mg ethyl cellulose ( $C_{20}H_{38}O_{11}$ ) were dispersed in 5 ml pure ethanol and were vigorously stirred for 30 min at room temperature. Afterwards, the solution temperature was raised to 80 °C and the stirring continued for 60 min. At this step, 5 mg  $\alpha$ -terpineol along with 5 drops of Triton X-100 were added to the solution and the stirring continued for 30 min. The prepared paste was deposited on FTO by doctor blade technique. Previously, The FTOs were covered with a blocking layer of dense TiO<sub>2</sub> to prevent electron recombination with electrolyte and dye molecules. After 15 min aging, the samples were put into an air furnace and sintered at 500 °C for 30 min. After cooling, the samples were soaked in 0.3 mM N179 dye (Dyesol Co., Australia) solution for 24 h at room temperature.

The solar cells were assembled by different counter electrodes. A Syrlin foil was used as a spacer, and the space between the two electrodes was filled with iodide-triiodide electrolyte (0.5 M lithium iodide (LiI), 0.05 M iodine (I<sub>2</sub>), and 0.05 M *tert*-butyl pyridine dissolved in

acetonitrile). Also, a control DSSC sample with a standard Pt counter electrode was prepared for comparison.

#### 2.4. Characterization

To investigate the crystal structure of prepared nanostructures, XRD diffraction patterns were attained by STOE STADI-P X-ray diffractometer. In order to better examine the structure and chemical bonding of the synthesized samples, FTIR studies were performed by Bruker-Vector 22 in the range of 400–4000  $\text{cm}^{-1}$ . The synthesized materials were mixed with KBr to form pellets. These pellets were placed inside the spectrometer and the transmission spectrum was measured. After extracting the FTIR data, the KBr spectrum was subtracted from the overall spectrum and the result was obtained in the form of a normalized curve. WITEC ALPHA300 RA-confocal Raman microscope with 532 nm excitation wavelength were used for structural and Raman analysis of synthesized samples. To inspect surface morphology of the synthesized counter electrodes, as well as the distribution of atomic elements, FESEM images and EDS analysis were prepared by Mira III Tescan field emission electron microscope with 15 kV operating voltage. A UV-Vis spectrophotometer (PerkinElmer Lambda 35) was used to determine the light absorption of the counter electrode samples. CV, Tafel, EIS, and Mott-Schottky electrochemical analyses were accomplished by potentiostat/galvanostat (EG&G model 273 A) in order to investigate the electrocatalytic activity and charge transfer properties of the manufactured samples. A 2450 Keithley source measure unit (SMU) with a solar simulator containing 350 W Xenon lamp equipped with AM1.5 filter (100 mW/cm<sup>2</sup>) was employed for I–V tracing. The PVE300 photovoltaic QE system Bentham was used for IPCE analysis.

#### 3. Results and discussion

Fig. 1 depicts the diffraction patterns for ZCO,  $WO_3$  and ZCO/ $WO_3$  samples. The peaks appearing for the ZCO sample at diffraction angles of 18.98, 31.33, 36.93, 38.52, 44.88, 55.64, 59.35 and 65.18°, respectively, represent the crystal planes (111), (220), (311), (222), (400), (422), (511) and (440) from the cubic phase of zinc cobalt oxide with Fd3m spatial symmetry. These peaks have a good coincidence with the standard card number 23-1390. According to the pattern, there are no



Fig. 1. XRD patterns for ZCO,  $WO_3$  and ZCO/ $WO_3$  synthesized nanostructures. At the bottom the XRD diffractions peaks are depicted for standard cards 23-1390 and 20-1324, which associated with ZCO and  $WO_3$  materials, respectively.

additional phases and impurity in ZCO and the substance is single-phase. Moreover, by employing Debye-Scherrer equation [35];

$$D = \frac{k \lambda}{\beta \cos(\theta)},\tag{1}$$

It is possible to estimate the crystallite size for this sample as an average of 28 nm. The sharpness of the peaks in the XRD patterns indicates the formation of larger crystals, which can be effective in the electron transport process to the surface of the counter electrode as well as its transfer to the electrolyte. The diffraction pattern of WO<sub>3</sub> sample shows peaks at angles of 23.30, 23.67, 24.27, 28.73, 33.33, 34.14, 35.62, 41.62, 47.21, 48.36, 49.80, 54.73, 55.76, 60.32, 62.36°, which are respectively associated with (001), (020), (200), (111), (021), (220), (121), (221), (002), (040), (140), (240), (420), (241) and (430) of the orthorhombic WO3 with PE spatial symmetry (standard card number 20-1324). The observed peaks show the good purity of the sample and the absence of secondary phases, which guarantees the quality of the material. Also, by using the Debye-Scherrer relation, it can be shown that the size of the crystals in this sample is around 19 nm, which is less than the ZCO sample. The reason for this is the broadening of the peaks, which of course can lead to the smaller crystallite size. As can be seen in the top pattern of Fig. 1, the ZCO/WO<sub>3</sub> composite includes peaks that can be considered a combination of the peaks of the separate phases of ZCO and WO<sub>3</sub>. It can be imagined with a proper approximation that the ZCO and WO<sub>3</sub> samples are well combined and have formed a hybrid structure. However, the reaction between ZCO and WO3 has not caused the creation of a new phase of other substances, and we are only dealing with a composite of the two materials.

The surface morphology of the synthesized counter electrodes is illustrated in Fig. 2. FESEM images acquired from ZCO, WO3 and ZCO/ WO3 samples, correspondingly represent the surface structure of different counter electrodes in Fig. 2a, b and c. The morphology of ZCO is composed of nanoparticles include small pores. However, WO3 consists of a more compact and uniform surface that does not have many pores. In Fig. 2c, the surface morphology of ZCO/WO3 shows a composite state of ZCO nanoparticles and WO3 nanostructures, which evokes a suitable involvement between ZCO and WO<sub>3</sub>. This layer can provide the characteristics of ZCO and WO3 at the same time. The distribution for atomic elements of ZCO/WO3 sample can be realized in Fig. 2d along with EDS mapping for different elements O, Co, Zn and W. As can be seen, this layer is a combination of different elements with atomic percentages of 57, 19, 11 and 13% for O, Co, Zn and W elements, which has a small oxygen vacancy compared to the stoichiometric state. The atomic distribution and also the morphology of the ZCO/WO<sub>3</sub> counter electrode along with the XRD diffraction patterns indicate the formation of a suitable layer of ZCO and WO3 combination, which can be used as an effective counter electrode in the DSSC.

In addition to the morphological and structural study of the synthesized samples, to inspect the formation of chemical bonding in the ZCO and WO<sub>3</sub> samples, FTIR spectra were prepared from them. Fig. 3a shows the FTIR spectra for the different counter electrodes. Both the samples have a broad peak in the range of  $3000-3700 \text{ cm}^{-1}$ , which indicates the stretching O-H bonds on the surfaces of the materials [36, 37]. The slight peaks can be observed at 1650  $\text{cm}^{-1}$ , which specifies H–O–H bending modes and the physical presence of water at the surface [38]. The peaks are witnessed at 553 and 686  $\text{cm}^{-1}$  in the ZCO sample represent the oscillation modes that related to Co-O and Zn-O bonding, respectively [39]. These two bonds can indicate the ZCO spinel phase in the counter electrodes [40]. Furthermore, in the WO<sub>3</sub> sample, the peaks appearing at 580 and 938 cm<sup>-1</sup>, respectively characterize the oscillation modes of W–O and W=O, which justify the presence of WO<sub>3</sub> [41]. The W=O peak is usually formed when the substance is hydrated [42]. The sharpness of FTIR peaks, along with the reference to the mentioned chemical bonds, is conducive to the formation of high-quality and pure crystalline structures of ZCO and WO3 materials, which are in good agreement with the XRD patterns.



Fig. 2. FESEM images for (a) ZCO, (b) WO<sub>3</sub>, and (c) ZCO/WO<sub>3</sub> samples. (d) The EDS atomic distribution and EDS mapping for the ZCO/WO<sub>3</sub> sample.



Fig. 3. (A) FTIR and (b) Raman spectrum for WO<sub>3</sub> and ZCO samples.

To further investigate the chemical bonds and structure of the synthesized samples, we used Raman analysis to complete the FTIR results. Fig. 3b shows the Raman spectra for the synthesized WO<sub>3</sub> and ZCO samples. As can be seen, the Raman peaks are very sharp, indicating the good crystalline quality of the synthesized samples. With more order of the crystal structure, the oscillating modes become more uniform and thus the peaks appear sharper. In WO<sub>3</sub> sample, four characteristic peaks are seen in the range of 200–1000 cm<sup>-1</sup>. The prominent and sharp peaks appearing in 717 and 808 cm<sup>-1</sup> are typical peaks of the WO<sub>3</sub> crystal structure, which usually represent the bridging oxygen stretching oscillations (O–W–O) [43]. In addition, two peaks at 274 and 328 cm<sup>-1</sup> are related to W-O bending vibrations [44]. Furthermore, five sharp peaks can be recognized in the ZCO sample. The peaks appearing in wavenumbers 183, 516 and 611  $\mbox{cm}^{-1}$  represent the  $F_{2g}$  phonon modes, which include Zn–O oscillations in the tetrahedral structure of the spinel ZCO [39]. The sharp peak of 688 is also related to Co-O stretching oscillations in the octahedral structure of ZCO [45]. The peak observed in 478  $cm^{-1}$  represents the  $E_g$  phonon modes for the ZCO spinel structure [46].

Fig. 4a shows the absorption spectra in the UV and visible range for  $WO_3$ , ZCO and ZCO/ $WO_3$  counter electrodes.  $WO_3$  and ZCO counter electrodes have characteristic peaks in the range of 300 and 400 nm that distinguish them from each other. However, the absorption spectrum of the ZCO/ $WO_3$  composite sample overlaps the two spectra of ZCO and  $WO_3$ , which indicates the combination of these two materials. The optical bandgap of the counter electrode samples can be measured according to the Tauc equation:

$$\alpha h \nu = A \left( h \nu - E_g \right)^n. \tag{2}$$

In this equation,  $\alpha$  is the absorption coefficient,  $h\nu$  denotes the photon energy, A is a constant,  $E_g$  represents the bandgap, and n is a constant number. n = 1/2 is used for direct semiconductors and n = 2 for indirect semiconductors. Fig. 4b shows the Tauc plots for the different synthesized counter electrodes. According to Tauc equation, when  $ah\nu = 0$ , the



Fig. 4. (A) Absorption spectra, and (b) Tauc plots for different synthesized counter electrodes.

photon energy will be equal to the semiconductor bandgap. Therefore, it is possible to calculate the bandgap by fitting the linear part of the Tauc plots. It should be noted that WO<sub>3</sub> is considered as an indirect semiconductor, so its plot is drawn with n = 2. We have used n = 1/2 to draw the ZCO/WO<sub>3</sub> Tauc plot. Beside the fact ZCO is a direct semiconductor, higher percentage of the ZCO/WO<sub>3</sub> counter electrode is made of ZCO, and the indirect bandgap of WO<sub>3</sub> is less obvious in it. The optical bandgap for all three counter electrodes is almost close to each other and is in the range of 2.71–2.83 eV. These results have been previously mentioned in other researches [47,48].

In order to investigate the electrochemical properties of different counter electrodes and determine their electrocatalytic and transport properties, CV, Tafel, EIS, and Mott-Schottky analyses were prepared. Fig. 5a shows the CV curves for ZCO, WO<sub>3</sub>, ZCO/WO<sub>3</sub> and standard Pt counter electrodes. In this analysis, a conventional 3-electrode electrochemical cell was used with WO<sub>3</sub>, ZCO/WO<sub>3</sub>, and Pt working electrode, Ag/AgCl reference electrode and a platinum wire as counter electrode. We also used the  $I^-/I_3^-$  electrolyte solution as the medium of redox reactions to illustrate the interactions occurred on the surface of the counter electrode in the DSSC. The applied potential performs in the range between -0.5 V and 1.5 V by the potentiostat with rate of 10 mV/ s. As Fig. 5a shows, the CV curves consist two reduction and oxidation peaks, which represent the reactions  $2I_2+2e^{-} \Leftrightarrow 2I_3^{-}$  and  $I_3^{-}+2e^{-} \Leftrightarrow 2I^{-}$ [49]. In the DSSC mechanism under light irradiation, the electrolyte species reduction process occurs on the surface of the counter electrode, so the half-reaction  $I_3^-+2e^-\rightarrow 2I^-$  is of special standing [50]. The faster and more intense the counter electrode can accomplish the reduction reaction, it actually provides more electrons for the regeneration the dye molecules and can increase the photocurrent. In this way, CV analysis can evaluate the electrocatalytic activity of the counter electrode with more current peak. In fact, with the increase of the peak related to  $I_3^-+2e^- \rightarrow 2I^-$  reaction, the electrocatalytic activity of the counter electrode increases, and it indicates the better performance of the solar cell. As can be seen from Fig. 5a, the half-reaction peak in WO<sub>3</sub> sample shows the lowest value, which indicates its low electrocatalytic activity. The ZCO sample has a larger peak, however, compared to the Pt sample, it is still less active. Although both WO<sub>3</sub> and ZCO counter electrodes alone have inferior electrocatalytic activity compared to Pt, the ZCO/WO<sub>3</sub> composite sample demonstrates higher peak than Pt, and this indicates the enhanced electrocatalytic activity.

Tafel analysis was prepared to better evaluate the electrocatalytic

activity of the synthesized counter electrodes. The semi-logarithmic plots of the anodic and cathodic current vs over-potential can be seen in Fig. 5b for different samples. The intersection of the diagrams in the Tafel zone with the  $\eta=0$  axis represents the exchange current (J\_0), which can estimate the electrocatalytic activity of the layers [51]. The higher exchange current leads to the more electrocatalytic activity. As can be seen in the figure, the exchange current for the ZCO/WO\_3 sample is higher than the ZCO, WO\_3 and even Pt sample, which indicates its better performance.

To determine the charge transfer properties of different counter electrode, electrochemical impedance spectroscopy (EIS) was achieved. The results of this analysis can be seen as Nyquist plots in Fig. 5c. We used a symmetrical structure of CE/electrolyte/CE in the setup of this analysis, in which a pair of counter electrodes is immersed in the Na<sub>2</sub>SO<sub>4</sub> electrolyte solution to eliminate the other effects. The frequency is modulated between 0.1 Hz and 20 kHz and a disturbance potential of 10 mV is introduced to the arrangement at each frequency point under dark condition and at room temperature. Normally, one semicircle should be appeared for each interface in Nyquist plot [52]. However, in all samples, only one semicircle is realized, which is due to the symmetrical structure, since both the electrodes are selected as the same material. The diameter of the semicircles represents the charge transfer resistance at counter electrode/electrolyte interface [53]. Also, the real part of the impedance at high frequencies that produce the points on the left side of the Nyquist plot is known as the series resistance (R<sub>S</sub>). This resistance is caused by the contact resistance, the series resistance of the solution and the FTO. As it is clear from Fig. 5c, the R<sub>S</sub> for all the samples is in the same range of 10–39  $\Omega$  cm<sup>2</sup>. The series resistance of the Pt sample is the lowest due to the better conductivity of the sample. The charge transfer resistance (R<sub>CT</sub>) for WO<sub>3</sub>, ZCO, ZCO/WO<sub>3</sub> and Pt samples is 565, 450, 235 and 357  $\Omega$  cm<sup>2</sup>, respectively. As it is realized, the pure WO<sub>3</sub> and ZCO samples show higher charge transfer resistance than Pt, but the charge transfer resistance for the composite ZCO/WO<sub>3</sub> electrode is significantly lower than that of Pt sample.

Lastly, to better investigate the electronic properties of the synthesized counter electrodes, we prepared Mott-Schottky analysis from them. The curves in Fig. 5(d–f) show the  $1/C^2$  in terms of applied potential for WO<sub>3</sub>, ZCO and ZCO/WO<sub>3</sub> counter electrodes. As can be seen, the slope of  $1/C^2$  curve in Fig. 5d related to WO<sub>3</sub> is positive, which indicates that this counter electrode is n-type semiconductor. WO<sub>3</sub> is considered an intrinsically n-type semiconductor, which usually arises



Fig. 5. (A) CV curves, (b) Tafel diagrams, (c) Nyquist plots, (d-f) Mott-Schottky analyses for different counter electrodes.

from oxygen vacancy defects [54]. Contrariwise, the  $1/C^2$  curve slope for the ZCO counter electrode is negative, which means that this semiconductor is p-type semiconductor. ZCO usually possess natural defects such as  $Zn_{Co}$  antisite, which are known to be the main source of p-type conductivity in this material [55]. The Mott-Schottky curve of the composite ZCO/WO<sub>3</sub> sample includes an inverted V-shape feature, which is reminiscent of a p-n junction. This behavior has also been observed before in p-n junctions [56]. In fact, the combination of p-type ZCO and n-type WO<sub>3</sub> produces a kind of p-n junction, which can even contribute to the better performance of the DSSC [30]. The charge separation in the counter electrode makes their transport more efficient. The carrier density can be approximated using the Mott-Schottky equation:

$$\frac{A^2}{C^2} = \frac{2}{\varepsilon \varepsilon_0 q N_d} \left( V - V_{FB} - \frac{k_b T}{e} \right).$$
(3)

In this equation, C is the space charge capacity in the semiconductor, A

stands for the surface area of the counter electrode, q is the elementary charge,  $N_d$  is carrier concentration,  $\epsilon\epsilon_0$  represents the semiconductor permittivity, V and  $V_{FB}$  are the applied and flat band potentials, k is Boltzmann's constant, and T is absolute temperature. The carrier density can be obtained by calculating the slope of the curves in Fig. 5d-f. For WO<sub>3</sub> counter electrode, the carrier density is approximately  $2.86 \times 10^{18}$  $\rm cm^{-3},$  while the carrier density for the ZCO sample is estimated to be  $9.14~\times~10^{17}~cm^{-3}.$  The carrier density in the ZCO/WO\_3 composite sample is similarly estimated by calculating two slopes in the linear region, which has a slight change compared to the single WO<sub>3</sub> and ZCO samples. In ZCO or WO<sub>3</sub> counter electrodes, holes or electrons must travel through the nanostructure semiconductor and react with electrolyte on the CE surface. While in the composite ZCO/WO3 CE, electrons and holes are separated faster, allowing electrons to proceed on the path with minimal recombination. Likewise, decreased recombination improved the charge transfer process at the CE/electrolyte interface, enhancing electrocatalytic activity [30]. In the semiconductor science





Fig. 5. (continued).

point of view, the holes that travel from the electrolyte to the CE can be injected into the CE and delivered into the circuit more easily.

The electrocatalytic activity of the ZCO/WO<sub>3</sub> counter electrode is much higher than that of the pure ZCO and WO<sub>3</sub> ones. This hybrid counter electrode has even better activity than the Pt sample. All four imperative CV. Tafel, EIS, and Mott-Schottky analyzes confirm the fact that this compound can improve the electrocatalytic properties. Moreover, the electrocatalytic activity trends have been well repeated in all the analyses, which validate the effectiveness of the composite structure. The reason for the improvement of electrocatalytic activity can be explained as follows; WO<sub>3</sub> layers have good transport properties due to their almost dense and uniform structure, but they do not have much porosity to have more contact surface with the electrolyte. On the contrary, the ZCO sample has a higher effective surface and can be effective in interacting with the electrolyte. However, both individual WO3 and ZCO samples do not have much chance to compete with the Pt counter electrode. The ZCO/WO<sub>3</sub> layer, on the other hand, includes the porous part of ZCO due to nanoparticles and also benefits from the good charge transport properties of WO<sub>3</sub>. In this sense, similar to hierarchical structures, ZCO nanoparticles are responsible for exchanging electrons with the electrolyte, and WO3 material is responsible for transporting



After the investigating the structural and electrochemical properties of different counter electrodes, the solar cells were characterized with assembled DSSCs based on various counter electrodes and TiO<sub>2</sub> photoanode. The results of their current-voltage characteristic curves can be seen in Fig. 6a. Also, the photovoltaic parameters are tabulated in Table 1. The DSSC assembled by ZCO/WO<sub>3</sub> counter electrode having open circuit voltage (V<sub>OC</sub>) of 691 mV, short circuit current (J<sub>SC</sub>) of 17.54 mA/cm<sup>2</sup> and filling factor (FF) of 64% shows the highest efficiency (PCE) of 7.76%. The Pt, ZCO, and WO<sub>3</sub> DSSCs rank next with efficiencies of 7.26, 5.57, and 3.25%, respectively. Considering the results and the

Table 1

Photovoltaic parameters for DSSCs fabricated by different counter electrodes extracted from I–V curves.

Solar cell	V <sub>OC</sub> (mV)	J <sub>SC</sub> (mA/cm <sup>2</sup> )	FF (%)	PCE (%)
WO <sub>3</sub> ZCO ZCO/WO <sub>3</sub> Pt	$\begin{array}{c} 591 \pm 9 \\ 647 \pm 26 \\ 691 \pm 5 \\ 683 \pm 3 \end{array}$	$\begin{array}{c} 8.89 \pm 0.08 \\ 13.88 \pm 0.07 \\ 17.54 \pm 0.10 \\ 16.87 \pm 0.06 \end{array}$	$\begin{array}{c} 62 \pm 0.7 \\ 62 \pm 0.7 \\ 64 \pm 0.8 \\ 63 \pm 0.7 \end{array}$	$\begin{array}{c} 3.26 \pm 0.08 \\ 5.57 \pm 0.19 \\ 7.76 \pm 0.09 \\ 7.26 \pm 0.11 \end{array}$



Fig. 6. (A) J-V characteristics, and (b) IPCE curves for different solar cells assembled with WO<sub>3</sub>, ZCO, ZCO/WO<sub>3</sub>, and Pt counter electrodes.

fact that the photoanodes are all the same, one can comprehend the fact that the effect of the counter electrode is impressive. The DSSC based on the ZCO/WO<sub>3</sub> counter electrode with the best electrocatalytic activity shows the highest efficiency. The significant effect of the counter electrode is mostly to increase the short circuit current. In actual fact, the ZCO/WO<sub>3</sub> counter electrode supplies electrons to the electrolyte with higher speed and intensity, so, the current passing through the cell increases. Furthermore, a slight improvement in FF and  $V_{OC}$  is observed, which can also be attributed to the better activity of the counter electrode. The ZCO/WO<sub>3</sub> counter electrode has significantly improved the solar cell performance compared to the pure samples by having better charge transport and transfer properties than the separate ZCO and WO<sub>3</sub> samples.

Fig. 6b shows the incident photon-to-current efficiency (IPCE) curves for different solar cells based on different counter electrodes in the range of 300–800 nm. The ZCO/WO<sub>3</sub> sample demonstrates a higher IPCE in all wavelengths than its competitors. In the range of 470–580 nm, where the IPCE diagram reaches its maximum value due to the absorption region of the N719 molecule [57], the ZCO/WO<sub>3</sub> solar cell has a quantum efficiency between 80 and 90%, which is 3–5% higher than that of the Pt electrode. This means the short circuit current for the ZCO/WO<sub>3</sub> sample is higher than that of the Pt sample and other samples. In addition, the integrated current density plots are depicted in Fig. 6b, using the equation:

$$J_{sc} = \int_{\lambda_1}^{\lambda_2} q.\Phi_{Ph}(\lambda).IPCE(\lambda) d\lambda$$
(4)

In this equation,  $\lambda_1$  and  $\lambda_2$  are the initial and final wavelengths of the spectrum,  $\Phi_{Ph}(\lambda)$  is the photon flux of AM1.5 spectrum, and q is the electron charge. The short circuit current obtained from the above equation and IPCE diagram for WO<sub>3</sub>, ZCO, ZCO/WO<sub>3</sub> and Pt samples is

equal to 8.43, 13.22, 17.15 and 16.41 mA/cm<sup>2</sup>, respectively. As can be seen, there is a good agreement between the short-circuit current extracted from the IPCE analysis and the J-V characteristic curve that previously observed.

Fig. 7 shows the Box and Whisker plots of the photovoltaic parameters for 60 DSSCs assembled with different counter electrodes. To examine the accurateness of the results and their reproducibility, we prepared 15 similar DSSC from each counter electrode sample. According to Fig. 7, the reproducibility of the results can be evaluated favorably. The PCE standard deviation in WO<sub>3</sub>, ZCO, ZCO/WO<sub>3</sub> and Pt samples is estimated to be 0.08, 0.19, 0.09, and 0.11%, respectively. It can be seen that by repeating the characterization of different DSSCs, almost the same results are obtained in each case.

#### 4. Conclusion

In this research, we fabricated ZnCo<sub>2</sub>O<sub>4</sub> (ZCO), WO<sub>3</sub> and ZCO/WO<sub>3</sub> layers using combustion, sol-gel and solid-state reaction methods and employed them as counter electrodes in DSSC structure. Examining the structural, morphological, and optical properties of the ZCO/WO3 sample shows that this layer is a composite of ZCO and WO<sub>3</sub>. Also, the investigation on the electrochemical properties shows that the electrocatalytic activity of the counter electrodes is improved by the formation of the composite. The CV, Tafel, EIS, and Mott-Schottky analyses confirm that the charge transfer properties and electrocatalytic activity of the ZCO/WO<sub>3</sub> sample are enhanced compared to their pure samples. The increase in electrocatalytic activity can be attributed to the hierarchical structure and taking advantage of the unique properties of ZCO/ WO<sub>3</sub> components. In this way, the charge transport and transfer properties were improved by the structures of WO3 and ZCO nanoparticles in the ZCO/WO3 counter electrode. Also, the DSSC based on ZCO/WO3 counter electrode attained the highest efficiency among all the other



Fig. 7. Box and Whisker plots for photovoltaic parameters achieved from 60 DSSCs.

competitors by obtaining an efficiency of 7.76%, which can promise to get rid of expensive Pt counter electrodes.

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#### CRediT authorship contribution statement

Raed H. Althomali: Methodology, Writing – original draft, Formal analysis. Ebraheem Abdu Musad Saleh: Formal analysis, Investigation, review & editing. Ramesh S. Bhat: Project administration, Conceptualization, review & editing. Shavan Askar: Formal analysis, Investigation, review & editing. I.B. Sapaev: Formal analysis, Investigation, review & editing. Mazin A.A. Najm: Formal analysis, Investigation, review & editing. Benien M. Ridha: Visualization, Data curation, review & editing. Ali H. Alsalamy: Visualization, Data curation, review & editing. Russual Riyadh: Validation, Data curation, review & editing.

#### Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Data availability

Data will be made available on request.

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